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Structural stability of one-dimensional long-period structures in the TiAl₃ compound

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Abstract

The cohesive energies of L1₂, D0₂₂, D0₂₃ and selected one-dimensional long-period structures (1D-LPSs) based on the L1₂ structure in the Ti–Al system for the TiAl₃ composition have been obtained by *ab initio* calculations using the Vienna *ab initio* simulation package. The 1D-LPSs are described within an Ising-like antiphase boundary (APB) model and we take into account both cell-external and cell-internal relaxations in the determination of their structural stability. Thus, the values of the APB energies are obtained in the ideal, distorted and fully relaxed structures. The results show that it is necessary to consider long-range interactions in order to obtain reliable values of the APB energies. We also relate these so-obtained APB energies to the energetic value of an isolated APB.

1. Introduction

The study of early-transition-metal (TM) trialuminides, TMAl₃, is of both technological and fundamental interest. The TMAl₃ compounds are attractive as potential structural materials for use in high-temperature environments or as thermally stable precipitates for developing so-called 'super alumalloys'.

Furthermore, it is important to emphasize that the crystal structure of these TM trialuminides depends on the location of the early TM in the periodic classification. The stable cubic L1₂ structure occurs for ScAl₃ only at room temperature and tetragonal D0₂₂ and D0₂₃ structures appear as the number of d electrons of the TM element increases. It is tempting therefore to relate the ordering tendency of such TM compounds to their electron to atom ratio [1]. Another way to analyse this ordering tendency is to note that the tetragonal D0₂₂ and D0₂₃ structures are derived from the cubic L1₂ by inserting (001) antiphase boundaries (APBs)

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every fcc cube along the (001) axis for the $D0_{22}$ structure and every two fcc cubes for the $D0_{23}$ one. For instance, if the interactions between the APBs are neglected, the (001) APB energy may be deduced directly from the energy difference between $L1_2$ and $D0_{22}$ structures:

$$E_{APB} = E_{D0_{22}} - E_{L1_2}.$$
 (1)

Considering the energy difference from $L1_2$ of the $D0_{23}$ structure, one obtains in this simple model

$$E_{APB} = 2(E_{D0_{23}} - E_{L1_2}).$$
⁽²⁾

In relations (1) and (2), the APB energy is defined per atom if the energies of L_{12} , D_{22} and D_{23} are expressed per atom.

Another interest in using the APB as an energetic criterion for phase stability is to relate the mechanical properties of these compounds to their structures since the energies involved in the formation of APBs are instrumental in most theories of the yield behaviour of ordered intermetallics.

But let us turn back to the study of the structural stability of these arrangements, which is the purpose of the present work. We can say that, if the antiphase energy is small, the system will be degenerated since several arrangements of APBs will compete with the L1₂ structure. For some alloys, such a behaviour may lead to the occurrence of one-dimensional long-period structures (1D-LPSs). This kind of chemical ordering has been discovered by Johansson and Linde [2] in the Au–Cu alloys and, more recently, 1D-LPSs have been observed in the Ti–Al system, more particularly for the Ti_{1+x}Al_{3(1-x)} alloys [3, 4].

The understanding of the thermodynamical behaviour of 1D-LPSs represents a great challenge; for the TiAl₃ compound, we propose to study the structural stability of series of 1D-LPSs based on the L1₂ structure by means of *ab initio* total-energy calculations. We shall show that the simple model leading to equations (1) and (2) is not sufficient to explain the relative stability of such 1D-LPSs and we shall propose discussing the relative stability in the framework of a more sophisticated APB Ising model. More particularly, we shall focus on the influence of both cell-external distortion and cell-internal atomic displacements of the atoms on the interaction parameters of the APB Ising model. Indeed a study performed by Amador *et al* [5] using the full-potential linear muffin tin method has shown how the relaxation effects influence the relative stability of the L1₂, D0₂₂ and D0₂₃ structures in the TiAl₃ compound.

The remainder of the paper is as follows. In section 2, the 1D-LPSs based on the $L1_2$ structure are briefly described. In section 3, we present the APB Ising model, which will be used further to analyse the structural stability of the 1D-LPSs. The *ab initio* calculations have been performed using the Vienna *ab initio* simulation package (VASP); the basis of this computer code and the conditions of the calculations are given in section 4. The results of the VASP calculations are presented in section 5. The determination of the interaction parameters of the APB Ising model is presented in section 6. Finally, section 7 summarizes the paper and gives the conclusions.

2. One-dimensional long-period structures

The 1D-LPSs we shall consider in the following are derived from the cubic L1₂-type structure A_3B (figure 1). L1₂ is ordered on the face-centred cubic lattice and consists, along the (001) cubic direction, of a stacking of pure A planes alternately with mixed AB planes. In the stacking sequence of pure and mixed planes, the translation [001] connects minority atoms in subsequent mixed layers. In contrast, in the D0₂₂ structure presented in figure 1 a translation of c/2 along the z axis connects minority atoms to majority atoms. The D0₂₂ structure may



Figure 1. L1₂, D0₂₂ and D0₂₃ 1D-LPSs.

be viewed as a periodic arrangement of (001) APBs in the L1₂ structure, the minimal length between the APBs being equal to the height of the L1₂ cell. D0₂₂ is denoted 11 in Zdanov's [6] notation or $\langle 1 \rangle$ in the notation of Fisher and Selke [7, 8]. $\langle 2 \rangle$ 1D-LPS (or D0₂₃) consists of a stacking of four L1₂ cubes with the same antiphase shift every two cubes. The $\langle M \rangle$ 1D-LPS, *M* being an integer, consists of a stacking of 2*M* L1₂ cubes with the same antiphase shift every *M* cubes. The 1D-LPSs can be characterized by the average domain size, usually denoted *M*, which is the ratio of the period and the number of domains or the number of L1₂ unit cells divided by the number of APBs in the 1D-LPS unit cell.

Along the *z* axis, the 1D-LPSs can be seen as Ising chains of spins + and –. Here + and – spins describe the two atomic species A or B in the (001) mixed planes. The L1₂ structure corresponds to the ferromagnetic + + + + + + state as the D0₂₂ structure corresponds to the antiferromagnetic + - + - + - + - state. An APB with respect to the L1₂ structure is like a magnetic domain wall separating two different ferromagnetic domains. Alternatively an APB with respect to the D0₂₂ structure is like a magnetic domains. With this description of the APBs, the 44 (or (4)) 1D-LPS which is + + + - - - possesses in its unit cell two APBs with respect to L1₂ (++ + +/- - - -/) and six APBs with respect to D0₂₂ (+/+/+/+-/-/-). Similarly the 211211 (or (21²)) which is + + - - + - in the spinlike description possesses in its unit cell six APBs with respect to L1₂ (+ +/-/+/- -/-/-). Let us remark that the (211) 1D-LPS is obtained from (4) by overturning alternate spins. Similarly (1) is obtained from L1₂ by overturning alternate spins, and (21) from (3); in this spin change (2) is not modified. More generally, the (21^j) 1D-LPS with M' = j + 2 is obtained from the (M) 1D-LPS with M = M' by overturning alternate spins.

When one considers 1D-LPSs as based on the D0₂₂ structure, it is more convenient to give the period of the superstructure in term of the size M' of the D0₂₂ antiphase domain still measured in units of the underlying L1₂ structure. M' is related to M by $M' = \frac{M}{M-1}$. $M' = \infty$ for the D0₂₂ structure, 1 for L1₂, 2 for D0₂₃, 3 for $\langle 21 \rangle$, 4 for $\langle 211 \rangle$ described previously and j + 2 for the $\langle 21^j \rangle$ 1D-LPS.

3. Antiphase boundary energy model

As proposed by Bak and Bruinsma [9] and more recently by Rosengaard and Skriver [10], the energy difference of a 1D-LPS from the underlying structure, either L1₂ or D0₂₂, may be mapped onto a one-dimensional effective Ising Hamiltonian in a field. In this description, the presence of an APB is represented by \uparrow with $\sigma = \frac{1}{2}$ while the absence is represented by \downarrow with $\sigma = -\frac{1}{2}$; the field is given by the APB energy, E_{APB} . The interactions between neighbouring APBs are given by I_n and $H_{n,m}$ for two-and three-body interactions respectively. The energy difference from the reference structure for a given 1D-LPS is

$$E_{S} - E_{\langle ref \rangle} = \frac{1}{N} \sum_{i} \left(\sigma_{i} + \frac{1}{2} \right) E_{APB} + \frac{1}{N} \sum_{j>i} I_{j-i} \left(\sigma_{i} + \frac{1}{2} \right) \left(\sigma_{j} + \frac{1}{2} \right) + \frac{1}{N} \sum_{j>i>k} H_{i-k,j-i} \left(\sigma_{i} + \frac{1}{2} \right) \left(\sigma_{j} + \frac{1}{2} \right) \left(\sigma_{k} + \frac{1}{2} \right)$$
(3)

where the sums are restricted to nearest-neighbour spins. In the following we shall restrict the three-body interaction to the first term $H_{1,1}$.

When taking the L1₂ structure as the reference, the energy difference from L1₂ of $\langle M \rangle$ 1D-LPSs, *M* being an integer, is

$$M(E_{\langle M \rangle} - E_{\langle \infty \rangle}) = E_{APB} + I_M + I_{2M} + \cdots$$
(4)

The APB energy with respect to L1₂, E_{APB} , is the limit of $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ when $M \to \infty$, but one can expect that expansion (4) converges rapidly. If this is the case, the value of $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ becomes constant from some small value of M and the APB energy with respect to the L1₂ structure is equal to this constant.

When taking the D0₂₂ structure as the reference, energy differences from D0₂₂ of the 1D-LPS $\langle 21^j \rangle$, *j* being an integer, are given by

$$M'(E_{(21^{j})} - E_{(1)}) = E'_{APB} + I'_{M'} + I'_{2M'} + \cdots$$
(5)

with M' = j + 2. The APB energy with respect to D0₂₂ is the limit of $M'(E_{(21j)} - E_{(1)})$ when $M' \rightarrow \infty$, and if expansion (5) converges rapidly the value of $M'(E_{(21j)} - E_{(1)})$ becomes constant from some small value of M' and the APB energy with respect to D0₂₂, E'_{APB} , is equal to this constant. One can show that the coefficients of the two expansions (4) and (5) are related. The connecting relation between the APB energies is

$$E'_{APB} = -E_{APB} - 2\sum_{M_0-1} I_{M_0-1} - 3H_{1,1}$$
(6)

where M_0 is the value of M or M' from which the interaction coefficients can be taken equal to zero.

The purpose of the following is to obtain, from *ab initio* total-energy calculations, the energy differences from L1₂ of series of 1D-LPSs $\langle M \rangle$ with M an integer and $\langle 21^j \rangle$. From these energy differences, the APB energy and the interaction parameters will be obtained. Moreover, we shall show that this determination depends on how we compute the energy differences.

4. Ab initio calculations

The calculations presented here were performed using the VASP, which has been described elsewhere [11–13]. All calculations were performed in the generalized gradient approximation (GGA) proposed by Perdew and Wang [14]. The electron–ion interaction is

described by ultrasoft pseudopotentials which allow the use of a moderate cut-off for the construction of the plane-wave basis even for TMs (222 eV for Ti). For the present calculation of ultrasoft pseudopotentials, the atomic reference configurations were $3p^6 4s^1 3d^3$ for Ti and $3s^2 3p^1$ for Al. For Ti, it is essential to include the 3p states as valence states in order to obtain correct lattice parameters. For Ti, the radii for the calculations of the augmentation functions are $R_{aug,l} = 2.20$, 2.00 and 2.49 au for l = 2, 1, 0, respectively; for Al, $R_{aug,l} = 2.36$ for l = 1 and 0. Partial core corrections were introduced to enable a proper treatment of the nonlinear dependence of the exchange–correlation functional on the charge density.

For the total-energy calculations of the Al₃Ti compound in the L1₂ structure (four atoms in the primitive cell) a 10 × 10 × 10 k-point mesh was chosen. For the 1D-LPS, the same k-point mesh along x and y axes was retained. Along the z axis, the number of k points was reduced as the length of the 1D-LPS increased. We saw in section 3 that the important result of the calculations is the energy difference of a given 1D-LPS from the L1₂ structure. To improve the precision of this difference, a L1₂ superstructure containing the same number of L1₂ unit cells as the considered 1D-LPS has been studied using the same number of k points and its cohesive energy calculated. For each 1D-LPS, the energy difference from L1₂ is calculated with respect to the corresponding L1₂ superstructure.

5. Results

5.1. Equilibrium structures

From L1₂ to D0₂₂, and to the other 1D-LPSs, symmetry elements are progressively lost so that relaxation degrees of freedom increase correspondingly. In the L1₂ structure, the energy is optimized with respect to the lattice parameter *a* only. In the D0₂₂ structure, the energy optimization is performed with respect to the lattice parameter *a* and with respect to the *c/a* ratio (tetragonal distortion of the lattice). Finally, in the D0₂₃ and other 1D-LPSs, the lattice parameter *a*, the *c/a* ratio (tetragonal distortion of the lattice) and additionally the cell-internal displacements of Ti and Al atoms are optimized.

The lattice parameters obtained in the volume optimization procedure are reported in figures 2 and 3 respectively for the $\langle M \rangle$ and $\langle 21^j \rangle$ 1D-LPSs. The tetragonality reported for one L1₂ unit cell is presented in figures 4 and 5 respectively for the $\langle M \rangle$ and $\langle 21^j \rangle$ 1D-LPSs. In the case of ideal structures, the convergence of the lattice parameters of the $\langle M \rangle$ 1D-LPSs to the value obtained in the L1₂ structure is fast. Similarly the convergence of the lattice parameters of the $\langle 21^j \rangle$ 1D-LPS to the lattice parameter of the ideal D0₂₂ is also fast. In contrast, for the distorted and fully relaxed phases $\langle M \rangle$ and $\langle 21^j \rangle$, the convergence becomes slower and one may observe that the lattice parameters *a* of the $\langle M \rangle$ 1D-LPSs differ from that of the L1₂ structure and that the lattice parameters *a* of the $\langle 21^j \rangle$ 1D-LPSs differ from that of the D0₂₂ structure.

The energy differences from L1₂ of all the 1D-LPSs studied in the present work are reported in table 1. In the ideal case, the ground state is L1₂. With distortion only, the D0₂₂ structure is the ground state. For the fully relaxed phases, the D0₂₃ structure is the most stable 1D-LPS; this result is unexpected because experimentally TiAl₃ crystallizes in the D0₂₂ structure [15]. However, our result is in perfect agreement with that obtained by Amador *et al* [5], who found that D0₂₃ is the ground state but by a very small margin. It must be noticed that the displacements of the atoms are small, and it is expected that these displacements will cancel with increasing temperature, explaining why D0₂₂ is observed experimentally at room temperature. One must also quote that the L1₂ structure has been observed experimentally under particular circumstances (see for example [16]). The values of the energy differences



Figure 2. Lattice parameters of the $\langle M \rangle$ 1D-LPSs. \Box , ideal structures; \triangle , distorted structures; O, fully relaxed structures.



Figure 3. Lattice parameters of the (21^j) 1D-LPSs. \Box , ideal structures; \triangle , distorted structures; O, fully relaxed structures.

from L1₂ of the series of 1D-LPS $\langle 21^j \rangle$ are very near that of D0₂₂ in the distorted case: this explains why 1D-LPSs with one and two bands have been observed experimentally in the Ti–Al system near the TiAl₃ composition by Miida *et al* [3] and by Loiseau *et al* [4].



Figure 4. Tetragonality of the $\langle M \rangle$ 1D-LPSs. \triangle , distorted structures; O, fully relaxed structures.



Figure 5. Tetragonality of the (21^j) 1D-LPSs. \triangle , distorted structures, O, fully relaxed structures.

5.2. Constrained structures

In the previous paragraph, we have seen that the lattice parameters *a* of the $\langle M \rangle$ 1D-LPSs differ from that of the L1₂ structure and that the lattice parameters *a* of the $\langle 21^j \rangle$ 1D-LPSs differ from that of the D0₂₂ structure. Therefore it seems difficult from the previous results to obtain the energy of an isolated APB either in the L1₂ or in the D0₂₂ structure. Indeed to obtain true information on an isolated APB it is necessary to perform calculations with a constant lattice parameter *a*, that means with constrained structures. Then, when studying the APBs in the

1D-LPS	Ideal	Distorted	Fully relaxed
$\langle 1 \rangle$	46.4	-25.0	-25.0
$\langle 2 \rangle$	10.8	-20.8	-33.2
(3)	5.5	-9.6	-21.9
$\langle 4 \rangle$	3.6	-4.5	-14.8
$\langle 5 \rangle$	3.5	-1.4	-10.5
$\langle 6 \rangle$	2.5	0.1	-8.7
$\langle 7 \rangle$	2.1	0.3	-7.5
$\langle 21 \rangle$	19.4	-18.9	-31.6
$\langle 21^2 \rangle$	25.8	-20.8	-30.9
$\langle 21^3 \rangle$	29.9	-20.8	-29.2
$\langle 21^4 \rangle$	32.5	-19.6	-28.4
$\langle 21^5 \rangle$	34.9	-19.9	-27.8

Table 1. Values of the energy differences from $L1_2$ of equilibrium 1D-LPSs in meV/atom.

Table 2. Values of the energy differences from L_{1_2} of 1D-LPSs constrained to the *a* lattice parameter of the equilibrium L_{1_2} structure in meV/atom.

1D-LPS	Distorted structures along z axis	Fully relaxed structures along z axis
$\langle 1 \rangle$	33.3	33.3
$\langle 2 \rangle$	2.0	-16.0
(3)	1.5	-13.0
$\langle 4 \rangle$	1.5	-10.7
$\langle 5 \rangle$	2.0	-8.2
$\langle 6 \rangle$	1.9	-6.7
(7)	1.4	-6.3

Table 3. Values of the energy differences from L1₂ of (21^j) 1D-LPSs constrained to the *a* lattice parameter of the equilibrium distorted D0₂₂ structure in meV/atom.

1D-LPS	Distorted structures along <i>z</i> axis	Fully relaxed structures along z axis
$\langle 1 \rangle$	-86.2	-86.2
$\langle 2 \rangle$	-78.6	-89.2
$\langle 21 \rangle$	-78.1	-89.0
$\langle 21^2 \rangle$	-81.4	-89.2
$\langle 21^3 \rangle$	-81.3	-88.9
$\langle 21^4 \rangle$	-82.6	-88.7
$\langle 21^5 \rangle$	-82.9	-88.4

L1₂ structure, the lattice constants in the *x* and *y* directions were fixed to the values obtained in the L1₂ structure. The c/a ratio along the *z* axis as well as the displacements in this direction were optimized. The results of such calculations are presented in table 2. Similarly, when studying the APBs in the D0₂₂ structure, the lattice constants in the *x* and *y* directions were fixed to the values obtained in the D0₂₂ structure. The c/a ratio along the *z* axis as well as the displacements in this direction were optimized. The results of such calculations are presented in table 3.



Figure 6. VASP-calculated values of the energy differences from L1₂ multiplied by *M* of $\langle M \rangle$ 1D-LPSs. \Box , ideal structures; Δ , distorted structures; O, fully relaxed structures. The crosshairs represent the values calculated with the APB model using the fitted APB energy and interaction parameters.

6. Application of the APB Ising model

6.1. Equilibrium structures

We shall first consider the equilibrium 1D-LPSs. In these conditions, the parameters of the APB Ising model will give indications of the relative stabilities of the 1D-LPSs. In the following we shall use two methods to derive the values of E_{APB} and E'_{APB} .

6.1.1. Method 1. In the APB model, the value of E_{APB} is theoretically obtained from the value of $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ when $M \to \infty$. Similarly the value of E'_{APB} is obtained from $M'(E_{(21^j)} - E_{(1)})$ when $M' \to \infty$. The energy differences from L1₂ of $\langle M \rangle$ 1D-LPSs, in the form $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$, are reported in figure 6. Likewise the $M'(E_{\langle 21 \rangle}^J - E_{\langle 1 \rangle})$ values for the $\langle 21^j \rangle$ 1D-LPSs are reported in figure 7. In each case, ideal, distorted and fully relaxed structures have been considered. In the ideal case, both $M(E_{(M)} - E_{(\infty)})$ and $M'(E_{(21^i)} - E_{(1)})$ values converge rapidly and become constant from M or M' equal to four. The value of E_{APB} is obtained as the mean value of $M(E_{(M)} - E_{(\infty)})$ for M = 4-7. Like wise, the value of E'_{APB} is obtained from the mean value of $M'(E_{(21^j)} - E_{(1)})$ for M' = 4-7. These values are reported in tables 4 and 5. In the case of distorted 1D-LPSs, the convergence is much slower, and it is only for the phases with M or M' equal to six and seven that both $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ and $M'(E_{(21^j)} - E_{(1)})$ are practically constant; the values of E_{APB} and E'_{APB} obtained with the $\langle M \rangle$ phases or with the (21^j) phases are reported in tables 4 and 5. For the fully relaxed phases, the situation is intermediate and one can consider that both $M(E_{(M)} - E_{(\infty)})$ and $M'(E_{(21^j)} - E_{(1)})$ are constant from M or M' equal to five. The corresponding values of E_{APB} and E'_{APB} are reported in tables 4 and 5. The method used above to obtain the APB energies will be called method 1 in the following.



Figure 7. VASP-calculated values of the energy differences from D0₂₂ multiplied by M' of $\langle 21^j \rangle$ 1D-LPSs. \Box , ideal structures; Δ , distorted structures; O, fully relaxed structures. The crosshairs represent the values calculated with the APB model using the fitted APB energy and interaction parameters.

Table 4. APB energies with respect to $L1_2$ of the equilibrium 1D-LPSs. The values are given in meV/atom.

	Ideal	Distorted	Fully relaxed
$E_{\rm D0_{22}} - E_{\rm L1_2}$	46	-25	-25
$2(E_{\rm D0_{23}} - E_{\rm L1_2})$	22	-42	-66
E_{APB} (method 1)	15	1	-52
E_{APB} (method 2)	16	-1	-55

Table 5. APB energies with respect to $\mathrm{D0}_{22}$ of the equilibrium 1D-LPSs. The values are given in meV/atom.

	Ideal	Distorted	Fully relaxed
$E_{L1_2} - E_{D0_{22}}$	-46	25	25
$2(E_{D0_{23}} - E_{D0_{22}})$	-36	4	-8
E_{APB} (method 1)	-82	34	-20
E_{APB} (method 2)	-83	39	-13

6.1.2. Method 2. However the strategy employed above to obtain the values of E_{APB} and E'_{APB} does not give any information on the interaction parameters between the APBs. To obtain this information, it is necessary to fit the values of the energy differences $E_{\langle M \rangle} - E_{\langle \infty \rangle}$ and $E_{\langle 21^{j} \rangle} - E_{\langle \infty \rangle}$ in a unique fit because we have shown before that the coefficients of the two expansions (4) and (5) are related. Therefore the values of $E_{\langle M \rangle} - E_{\langle \infty \rangle}$ and $E_{\langle 21^{j} \rangle} - E_{\langle \infty \rangle}$ and expression (3) to obtain the parameters of the APB Ising model. Note that only the first three-spin interaction term has be taken into account, let us say $H_{1,1}$.

Table 6. Fitted parameters of the APB Ising model with respect to $L1_2$. The values are given in meV/atom.

	Ideal	Distorted	Fully relaxed
E_{APB}	15.3	-0.9	-55.2
I_1	18.7	41.4	38.2
I_2	6.3	-21.6	-5.6
I_3	1.3	-24.2	-7.3
I_4	0	-19.1	-5.6
I_5	0	-11.1	0
$H_{1,1}$	4.9	9.7	9.7
σ	0.1	0.3	0.2

tests have been performed in which the number of effective pair interaction parameters has been modified. In each case, the standard deviation has been calculated; the goal is to obtain the best fit with the smallest number of parameters. In the ideal case, besides E_{APB} and $H_{1,1}$, it is necessary to introduce three more parameters, let us say I_1 , I_2 and I_3 . The fit of the values of $E_{\langle M \rangle} - E_{\langle \infty \rangle}$ and $E_{\langle 21^{J} \rangle} - E_{\langle \infty \rangle}$ is excellent. In the distorted case, seven parameters were necessary to obtain a good fit, E_{APB} , $H_{1,1}$ and I_1 – I_5 . For the fully relaxed structures, six parameters, E_{APB} , $H_{1,1}$ and I_1 – I_4 , were necessary to obtain the best fit. As expected, these numbers of parameters are in each case the same as predicted from inspection of figures 6 and 7. All the parameter values as well as the standard deviation are reported in table 6. The calculated parameters are those relative to the APB model with respect to L1₂.

The values of $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ and $M'(E_{\langle 21^j \rangle} - E_{\langle 1 \rangle})$ in the ideal, distorted and fully relaxed cases, recalculated with the three sets of parameters deduced from the fitting procedure, are reported in figures 4(a) and (b). In the ideal case, the representation is excellent. For the distorted and fully relaxed 1D-LPSs, the representation is less satisfying for *M* or *M'* equal to six or seven. This point will be explained further. The fitting procedure will be called in the following method 2.

6.1.3. Discussion on the APB model parameters. From inspection of the values of the APB interaction parameters (table 6), one observes that the first-nearest-neighbour interaction parameters are always strongly positive, showing a repulsion between first-nearest-neighbour APBs. The other interaction parameters are either positive and negative.

The values of E_{APB} and E'_{APB} obtained by method 2 are reported in tables 4 and 5 respectively, where they may be compared with the values obtained in method 1. In the ideal case, the difference between the values of E_{APB} and E'_{APB} obtained by method 1 and by method 2 is well within the accuracy of the calculations. In the distorted and fully relaxed cases some differences between the APB energies obtained by method 1 or method 2 appear. Let us recall that in method 1 E_{APB} and E'_{APB} are obtained independently for the $\langle M \rangle$ 1D-LPSs and for the $\langle 21^{j} \rangle$ ones, while in method 2 the two sets of values are treated in the same model. The observed differences are due to these different treatments. On the other hand, it must be mentioned that the difference is a maximum of 7 meV, which is very small when looking at the precision of the cohesive energy determination and at the fact that the energy differences are multiplied by *M* or *M'* in figures 6 and 7 and in tables 4 and 5.

Let us now discuss the values of E_{APB} and E'_{APB} energies. In the ideal case, E_{APB} is positive, showing that APB formation in L1₂ is not energetically favoured. In contrast, E'_{APB} is negative, showing that the formation of APBs in the ideal D0₂₂ structure is strongly energetically favoured.



Figure 8. VASP-calculated values of the energy differences from L1₂ multiplied by *M* of $\langle M \rangle$ 1D-LPSs whose lattice parameter, *a*, has been constrained to that of the L1₂ structure. \triangle , distorted structures; **O**, fully relaxed structures.

In the distorted case, E_{APB} is negative but very small in absolute value. If one looks at the values obtained for $\langle 6 \rangle$ and $\langle 7 \rangle$ 1D-LPSs, the APB energy with respect to L1₂ is slightly positive; the formation of APBs in the L1₂ structure to give 1D-LPSs of type $\langle M \rangle$ (*M* being an integer) is not energetically favoured. On the other hand, E'_{APB} is clearly positive: the creation of APBs in the D0₂₂ structure is not favourable energetically, and therefore the D0₂₂ structure is the stable one.

In the fully relaxed case, both E_{APB} and E'_{APB} are negative. Therefore, neither L1₂ nor D0₂₂ is the most stable structure, and indeed D0₂₃ is the ground state.

The values of E_{APB} and E'_{APB} , if they were obtained from the energy differences from L1₂ of D0₂₂ or of D0₂₃ structures, are reported in the two first rows of tables 4 and 5. Inspection of the values shows that, if the correct sign is almost always obtained, the values can differ by twice as much. Therefore, it is absolutely necessary to take into account the interactions between the APBs in order to obtain the APB energies. Moreover, it is necessary to perform *ab initio* calculations for a large number of 1D-LPSs to test the convergence of the expansion used in the APB model.

6.2. Constrained structures

Let us now consider the constrained 1D-LPSs. In this case, the values obtained for E_{APB} and E'_{APB} can be considered as those of isolated APBs either in L1₂ or in the D0₂₂ structures.

The values of $M(E_{\langle M \rangle} - E_{\langle \infty \rangle})$ obtained for the $\langle M \rangle$ 1D-LPSs constrained to have the same parameter as the equilibrium L1₂ structure are reported in figure 8. The values are practically constant from M = 5. The derived APB energies are reported in table 7.

The values of $M'(E_{\langle 21^j \rangle} - E_{\langle 1 \rangle})$ obtained for the $\langle 21^j \rangle$ 1D-LPSs constrained to have the same parameter as the equilibrium D0₂₂ structure are reported in figure 9. As above, these values are practically constant from M' = 5. The derived APB energies are reported in table 8.



Figure 9. VASP-calculated values of the energy differences from D0₂₂ multiplied by M' of $\langle 21^j \rangle$ 1D-LPSs whose lattice parameter, a, has been constrained to that of the D0₂₂ structure. \triangle , distorted structures; O, fully relaxed structures.

Table 7. APB energies with respect to $L1_2$ of the constrained $\langle M \rangle$ 1D-LPSs for the $L1_2$ lattice parameter *a*. The values are given in meV/atom.

	Distorted structures along <i>z</i> axis	Fully relaxed structures along <i>z</i> axis
$E_{\rm D0_{22}} - E_{\rm L1_2}$	33	33
$2(E_{D0_{23}} - E_{L1_2})$	4	-64
E_{APB} (method 1)	10	-42
$\gamma_{APB} \ (mJ \ m^{-2})$	40	170

Table 8. APB energies with respect to $D0_{22}$ for constrained 1D-LPSs at the lattice parameter *a* of the equilibrium $D0_{22}$ structure. The values are given in meV/atom.

	Distorted structures along <i>z</i> axis	Fully relaxed structures along z axis
$E_{\rm L1_2} - E_{\rm D0_{22}}$	86	86
$2(E_{D0_{23}} - E_{D0_{22}})$	30	-12
E_{APB} (method 1)	23	-15
$\gamma_{APB} \ (mJ \ m^{-2})$	98	65

As in section 6.1, we observe that the APB energies cannot be obtained solely from the energy differences from $L1_2$ of the $D0_{22}$ and $D0_{23}$ structures.

We must remark that method 2 indicated previously (section 6.1.2) has not been used because its application would necessitate the calculations of the (21^j) energies at the L1₂

lattice parameter and $\langle M \rangle$ energies at the D0₂₂ lattice parameter to obtain the interaction coefficients with sufficient precision.

The values of the APB energies reported in tables 7 and 8 can be considered as the energy of an APB isolated either in the L_{12} or in the D_{22} structure. These values reported to the area unit are given by

$$\gamma/L1_2 = \frac{E_{APB}}{a_{L1_2}^2/4}$$
 and $\gamma/D0_{22} = \frac{E'_{APB}}{a_{D0_{22}}^2/4}.$ (7)

The factor of four has been introduced because the APB energies given in tables 7 and 8 are expressed per atom. a is the lattice parameter along x and y axes of either the L1₂ or the distorted D0₂₂ structure.

In the approach developed in section 6.1, we obtained the energetic parameters of an APB Ising model for the purpose of describing the behaviour of the $L1_2$, $D0_{22}$, $D0_{23}$ and other 1D-LPSs. In this section we have obtained the energy of an isolated APB. Although these approaches are different, the comparison of the APB energies obtained with the equilibrium 1D-LPSs and with the constrained ones indicates only small differences.

7. Summary and conclusions

Some years ago Paxton [17] showed that it is possible to bring together the theory of alloys, as seen from the density functional point of view, and practical problems in physical metallurgy. In the present work, we have employed such a strategy to obtain APB energies from *ab initio* calculations of cohesive energies. Indeed the cohesive energies of two series of 1D-LPSs, let us say $\langle M \rangle$ with M an integer and $\langle 21^j \rangle$, were obtained with a code based on the density functional theory in the GGA. The calculations were performed for rather large values of M or M' (up to M or M' = 7). More, these calculations were performed for ideal, distorted and fully relaxed structures. To improve the energy difference from $L1_2$ of the 1D-LPSs, $L1_2$ superstructures have been built and their cohesive energies calculated with the same number of k points as used for the corresponding 1D-LPS. The results have been discussed in the framework of an APB Ising model, whose parameters have been obtained. These calculations have shown that, in the case of ideal structures, APB formation is not energetically favoured in the L_{12} structure while it is energetically favoured in the D_{22} structure. In the distorted case, the APB energy is very small in the L_{12} structure, while it is strongly positive in the D_{022} structure. This supports the fact that $L1_2$ can be maintained in a metastable state and that $D0_{22}$ is the most stable structure at ordinary temperature. In the fully relaxed situation, which can be considered as the situation at very low temperature, the APB energies with respect to L_{1_2} and to $D0_{22}$ are both negative, showing that neither $L1_2$ nor $D0_{22}$ is the ground state; actually at T = 0 K the ground state is the D0₂₃ structure.

We have also proved, in this work concerning the $TiAl_3$ compound, that the energy differences from $L1_2$ of $D0_{22}$ and $D0_{23}$ structures solely are not sufficient to derive APB energies with a good precision. Moreover, we have also shown that, in the $TiAl_3$ compound, all the relaxation effects, cell-external distortion and cell-internal displacements of the atoms, must be taken into account in order to obtain correct values of the APB energies.

Finally we have shown that the energy of an isolated APB obtained with constrained 1D-LPSs is not very different from that obtained with the APB Ising model applied to the equilibrium 1D-LPSs.

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